

PATENT ABSTRACTS OF JAPAN

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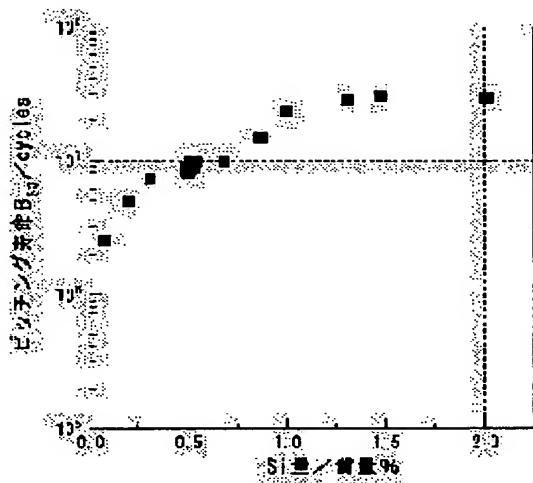
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HIRAOKA KAZUHIKO**(54) STEEL FOR INDUCTION HARDENING****(57)Abstract:**

PROBLEM TO BE SOLVED: To provide steel for induction hardening which has excellent wear resistance and pitting resistance.

SOLUTION: The steel for induction hardening contains alloy elements of, by mass, 0.30 to 0.50% C, 0.30 to 1.50% Si and 0.30 to 1.50% Mn, and the balance Fe with inevitable impurities, and is applied to parts having excellent wear resistance and an excellent pitting resisting life as shown in the figure.

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CLAIMS

[Claim(s)]

[Claim 1] Steel for induction hardening applied to the components with which the content of an alloy element exceeded [% / mass] in the remainder by consisting of Fe and an unescapable impurity at abrasion resistance and pitching-proof nature C:0.30 – 0.50%, Si:0.30-1.50%, and Mn:0.30-1.50%.

[Claim 2] Steel for induction hardening containing one sort which was chosen from nickel:0.03–3.00%, Cr:0.05–0.30%, and Mo:0.01–1.50% by mass % in addition to the alloy element according to claim 1, or two sorts or more applied to the components excellent in abrasion resistance and pitching-proof nature.

[Claim 3] Steel for induction hardening containing one sort which was chosen by mass % from V:0.05 – 0.50%, B:0.0005 – 0.0050%, Ti:0.02–0.05%, and Nb:0.005–0.200% in addition to the alloy element according to claim 1 or 2, or two sorts or more applied to the components excellent in abrasion resistance and pitching-proof nature.

[Claim 4] Steel for induction hardening which contains aluminum:0.005–0.050% and N:<=0.015% by mass % in any 1 term of claims 1–3 in addition to the alloy element of a publication and which is applied to the components excellent in abrasion resistance and pitching-proof nature.

[Claim 5] Steel for induction hardening containing one sort which was chosen as any 1 term of claims 1–4 by mass % from S:0.003 – 0.020%, Pb:0.03–0.20%, Bi:0.03–0.15%, and calcium:0.0003–0.0050% in addition to the alloy element of a publication, or two sorts or more applied to the components excellent in abrasion resistance and pitching-proof nature.

[Claim 6] Steel for induction hardening applied to the components excellent in abrasion resistance given in any 1 term of claims 1–5 whose Vickers hardness numbers at the time of annealing at 300 degrees C after induction hardening processing are 500 or more HV, and pitching-proof nature.

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DETAILED DESCRIPTION

[Detailed Description of the Invention]

[0001]

[Field of the Invention] This invention relates to the steel for induction hardening which obtains the high degree-of-hardness components excellent in abrasion resistance and pitching-proof nature.

[0002]

[Description of the Prior Art] In the steering components and shaft of an automobile, it is JIS until now. The coal mine was used while being represented by SCM440 and S45C. However, since the load concerning a steering etc. is becoming large in recent years, a demand to abrasion resistance and pitching-proof nature is increasing increasingly. Although the abrasion resistance in an elevated temperature and pitching-proof nature are needed in order that a surface degree of hardness may decrease by annealing in becoming hot environments especially during use, the present condition is that the demand is not attained.

[0003]

[Problem(s) to be Solved by the Invention] The place which this invention was made against the background of the above situations, and is made into the purpose of this invention is offering the steel for induction hardening excellent in abrasion resistance and pitching-proof nature.

[0004]

[Means for Solving the Problem] In order to raise abrasion resistance and pitching-proof nature, it is effective to raise resistance to temper softening. Therefore, it is JIS about Si said to raise resistance to temper softening. When it processed on the same induction hardening conditions as JIS steel by carrying out increase-in-quantity addition rather than SCM440 and S45C, it was possible to have raised resistance to temper softening rather than JIS steel, it found out that abrasion resistance and pitching-proof nature were improved greatly, and this invention was obtained.

[0005] That is, the means of this invention for solving the above-mentioned technical problem is steel for induction hardening applied to the components the content of an alloy element excelled [components] in invention of claim 1 at abrasion resistance and pitching-proof nature by the remainder consisting of Fe and an unescapable impurity by mass % C:0.30 – 0.50%, Si:0.30–1.50%, and Mn:0.30–1.50%.

[0006] It is the steel for induction hardening applied to the components containing one sort which is mass % and was chosen from nickel:0.03–3.00%, Cr:0.05–0.30%, and Mo:0.01–1.50% in addition to the alloy element of the means of claim 1 in invention of claim 2, or two sorts or more which were excellent at abrasion resistance and pitching-proof nature.

[0007] It is the steel for induction hardening applied to the components containing one sort which is mass % and was chosen in addition to the alloy element of the means of claims 1 or 2 in invention of claim 3 from V:0.05 – 0.50%, B:0.0005 – 0.0050%, Ti:0.02–0.05%, and Nb:0.005–0.200%, or two sorts or more which were excellent at abrasion resistance and pitching-proof nature.

[0008] It is the steel for induction hardening applied to the components which are mass %s and contain aluminum:0.005–0.050% and N:<=0.015% in invention of claim 4 in addition to the alloy element of the means of any 1 term of claims 1–3, and which were excellent at abrasion resistance and pitching-proof nature.

[0009] It is the steel for induction hardening applied to the components containing one sort which is mass % and was chosen in invention of claim 5 in addition to the alloy element of the means of any 1 term of claim –4 from S:0.003 – 0.020%, Pb:0.03–0.20%, Bi:0.03–0.15%, and calcium:0.0003–0.0050%, or two sorts or more which were excellent at abrasion resistance and pitching-proof nature.

[0010] In invention of claim 6, the Vickers hardness number at the time of annealing at 300 degrees C offers the steel for induction hardening in the means of any 1 term of claims 1–5 which are 500 or more HV applied to the components excellent in abrasion resistance and pitching-proof nature after induction hardening processing.

[0011] The reason for limitation of the alloy element in this invention is explained below. In addition, mass % shows %.

[0012] C:0.30 – 0.50%C is an element indispensable when securing the reinforcement of steel, and the content determines the hardness after induction hardening. So, in this invention, the minimum of C content was made into 0.30%, and hardness is secured. However, since the evil of reducing degradation and machinability of toughness will be brought about if there are too many the contents, an upper limit is made into 0.50%.

[0013] Si: 0.30 – 1.50%Si is an element which has an important role in this invention, dissolves in a matrix, and raises a tempering softening property by controlling a pearlite transformation. When the content is less than 0.30%, the effectiveness is not fully acquired, but since toughness and machinability will be reduced if many [too], an upper limit is made into 1.50%.

[0014] Mn: 0.30 – 1.50%Mn raises the hot-working nature of steel, and secures hardenability. When the content is less than 0.30%, the effectiveness is not fully acquired, but since machinability and cold forging nature will be degraded if many [too], an upper limit is made into 1.50%.

[0015] nickel: 0.03 – 3.00%nickel raises hardenability and makes the reinforcement in crystal grain increase. When the content is less than 0.03%, the effectiveness is not fully acquired, but since the grain boundary segregation of P will be promoted if many [too], an upper limit is made into 3.00%.

[0016] Cr: 0.05–0.30%Cr contributes to improvement in hardenability. When the content is less than 0.05%, the effectiveness is not fully

acquired, but since concentration into carbide will take place at the time of spheroidizing, carbide will remain at the time of induction hardening and uniform hardening hardness will not be obtained if many [too], an upper limit is made into 0.30%.

[0017] Mo: Mo contributes to improvement in Hardenability 0.01 to 1.50%. When the content is less than 0.01%, the effectiveness is not fully acquired, but since workability will be degraded if many [too], an upper limit is made into 1.50%.

[0018] V: 0.05 – 0.50%V contributes to grain refining, and raises toughness. When the content is less than 0.05%, the effectiveness is not fully acquired, but since a mechanical property will be degraded if many [too], an upper limit is made into 0.50%.

[0019] B: 0.0005 – 0.0050%B is an element which strengthens grain boundary reinforcement, and raises toughness, and raises Hardenability. When the content is less than 0.0005%, the effectiveness is not fully acquired, but since Hardenability will be conversely reduced if many [too], an upper limit is made into 0.0050%.

[0020] Ti: It is the element which Ti fixes free-N in steel 0.02 to 0.05%, and raises the effectiveness to the Hardenability of B, and raises toughness. In order that the amount of N in steel may specify it as 0.015% or less even if the effectiveness is not fully acquired but exceeds 0.05% when the content is less than 0.02%, since the effectiveness is saturated, it makes an upper limit 0.05%.

[0021] Nb: 0.005–0.200%Nb is an element which controls coarsening and raises toughness. When the content is less than 0.005%, the effectiveness is not fully acquired, but if many [too], in order to carry out precipitation hardening of the ferrite ground by carbon nitride generation, since deformation resistance is made to increase, an upper limit is made into 0.200%.

[0022] aluminum: 0.005 – 0.050%aluminum is an element required as deoxidation material, and it is the element which combines with N, serves as AlN and controls big and rough-ization of crystal grain. Since the effectiveness is not fully acquired, but an alumina system oxide will increase if many [too] and a fatigue property and workability are reduced when the content is less than 0.005%, an upper limit is made into 0.050%.

[0023] N: Since TiN will increase and <=0.015%N will have a bad influence on a fatigue property if it is contained exceeding 0.015%, it makes an upper limit 0.015%.

[0024] Each of S:0.003 – 0.020%, Pb:0.03–0.20%, Bi:0.03–0.15%, calcium:0.0003% – 0.0050%S, and Pb(s), Bi(s) and calcium are elements which raise machinability. When there are too few the contents, the effectiveness is not fully acquired, but since it has a bad influence on a property too as many, the range is determined.

[0025]

[Embodiment of the Invention] The steel of the gestalt of operation of this invention is shown in Table 1 and 2, drawing 1 – 4, and a list through an example. In Table 1, as for the gestalt of operation of claim 3, and the invention steel 4, the invention steel 1 shows [the gestalt of operation of claim 1, and the invention steel 2] the steel of the gestalt of operation of claim 5 for the gestalt of operation of claim 2, and the invention steel 3, as for the gestalt of operation of claim 4, and the invention steel 5–15.

[0026]

[Table 1]

(質量%)

	C	Si	Mn	S	Cr	Mo	Ti	B	Al	N
発明鋼	1 0.43	0.49	0.59	0.002	0.03	—	—	—	—	0.0053
	2 0.47	0.52	0.62	0.001	0.14	0.02	—	—	—	0.0059
	3 0.44	0.47	0.60	0.002	0.17	0.03	0.033	0.0014	—	0.0065
	4 0.45	0.49	0.59	0.002	0.15	0.03	0.028	0.0012	0.032	0.0065
	5 0.44	0.31	0.61	0.012	0.14	0.02	0.031	0.0013	0.033	0.0062
	6 0.43	0.51	0.60	0.011	0.14	0.03	0.029	0.0014	0.029	0.0049
	7 0.46	0.69	0.60	0.014	0.16	0.02	0.029	0.0012	0.028	0.0067
	8 0.47	0.88	0.58	0.009	0.17	0.02	0.033	0.0009	0.032	0.0056
	9 0.43	1.01	0.61	0.011	0.14	0.02	0.029	0.0012	0.030	0.0065
	10 0.45	1.32	0.62	0.014	0.15	0.03	0.030	0.0011	0.031	0.0055
	11 0.45	1.49	0.62	0.013	0.13	0.03	0.032	0.0013	0.033	0.0059
	12 0.46	0.53	0.60	0.012	0.28	0.02	0.028	0.0014	0.032	0.0049
	13 0.43	0.51	0.60	0.011	0.13	1.01	0.030	0.0012	0.029	0.0068
	14 0.44	0.49	0.58	0.014	0.16	0.04	0.033	0.0045	0.028	0.0065
	15 0.45	0.50	0.63	0.011	0.18	0.02	0.031	0.0014	0.031	0.0123
比較鋼	1 0.44	0.07	0.59	0.011	0.17	0.02	0.031	0.0011	0.033	0.0055
	2 0.46	0.20	0.60	0.014	0.16	0.02	0.033	0.0018	0.032	0.0068
	3 0.46	2.03	0.58	0.012	0.18	0.03	0.028	0.0013	0.028	0.0063
	4 0.45	0.54	0.62	0.013	0.52	0.02	0.029	0.0013	0.029	0.0060
	5 0.42	0.49	0.63	0.009	0.16	0.02	0.029	0.0099	0.029	0.0057
	6 0.47	0.52	0.61	0.012	0.14	0.02	0.030	0.0012	0.030	0.0211

[0027]

[Example] After ingotting the steel of the chemical composition shown in Table 1 with 100kg vacuum melting furnace, hot forging was carried out at 1150 degrees C, and the material with a diameter of 32mm was manufactured. Subsequently, after normalizing to said each material, as for this roller pitching test piece 1 that created the roller pitching test piece 1 shown in drawing 1, the diameter of a central major diameter has the dimension 26mm and whose die length are 28mm and whose die length the diameter of the narrow diameter portion of both sides is 22mm, and is 51mm. And after performing induction hardening processing which carries out a postscript to each roller pitching test piece 1, the roller pitching trial was performed.

[0028] Induction hardening processing was carried out on condition that the output of 40kW, the frequency of 200kHz, the electrical

potential difference of 9.5kV, current 3.2A, and coil passing speed 7 mm/sec, and performed tempering processing at 180 degrees C. [0029] Here, the principle of a roller pitching trial is shown in drawing 2. That is, in quest of 2, i.e., the number of rotations which is a life until it makes 3334Ns /carry out high-speed rotation under the planar pressure of 2 mm and pitching occurs, pitching-proof nature was evaluated for the roller pitching test piece 1 (small roller: quality of the material steel of Table 1), and the partner material 2 (large roller: 420 about JIS SCM) 340 kgf(s)/mm. In addition, the difference of the peripheral speed of the roller pitching test piece 1 (small roller) and the partner material 2 (large roller), i.e., a slip ratio, is 40% in this case. And after examining for five roller pitching test pieces 1 about a monograph affair, respectively, the Weibull plot was performed and pitching-proof nature was evaluated in quest of the breakage probability life (B50 life) 50%. Furthermore, the hardness in 100-micrometer location was measured under the 300-degree C high temperature service from the front face of a hard facing layer cross section. The result of induction hardening material is shown in Table 2 and drawing 3, and 4. According to drawing 3, the hardness in 300 degrees C is going up as the content of Si is made [many], and it is over 500HV by 0.3% or more of addition. In addition, the effectiveness is saturated as for 1.5% or more in the content of Si. Moreover, the effectiveness will be saturated, if the pitching life B50 is improving and a content is made 1.5% or more as the content of Si is made [many] according to drawing 4.

[0030]

[Table 2]

	300°C高温硬さ (HV)	ローラーピッティング 寿命 B ₅₀
発明鋼	1 502	7.8 × 10 ⁶
	2 514	8.4 × 10 ⁶
	3 510	8.2 × 10 ⁶
	4 513	8.0 × 10 ⁶
	5 501	7.3 × 10 ⁶
	6 512	8.5 × 10 ⁶
	7 521	9.6 × 10 ⁶
	8 631	1.5 × 10 ⁷
	9 639	2.3 × 10 ⁷
	10 547	2.8 × 10 ⁷
	11 554	3.0 × 10 ⁷
	12 515	6.8 × 10 ⁶
	13 523	9.6 × 10 ⁶
	14 519	8.7 × 10 ⁶
	15 511	7.9 × 10 ⁶
比較鋼	1 460	2.5 × 10 ⁶
	2 487	5.0 × 10 ⁶
	3 556	2.9 × 10 ⁷
	4 521	9.8 × 10 ⁶
	5 514	8.3 × 10 ⁶
	6 510	8.6 × 10 ⁶

[0031]

[Effect of the Invention] As explained above, this invention steel can make resistance to temper softening increase by adding Si so much, consequently is hard at an elevated temperature, and is the steel for induction hardening applicable to the components which have the property which was further excellent at pitching-proof nature.

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DESCRIPTION OF DRAWINGS

[Brief Description of the Drawings]

[Drawing 1] It is the side elevation showing the outline of the roller pitching test piece 1.

[Drawing 2] It is the perspective view showing the principle of a roller pitching trial.

[Drawing 3] It is the graph which shows relation with an amount [of Si], and a hardness of 300 degrees C.

[Drawing 4] It is the relation between the amount of Si, and a pitching life, and is the graph which shows the roller pitching test result of induction hardening material.

[Description of Notations]

1 Roller Pitching Test Piece

2 Partner Material

[Translation done.]

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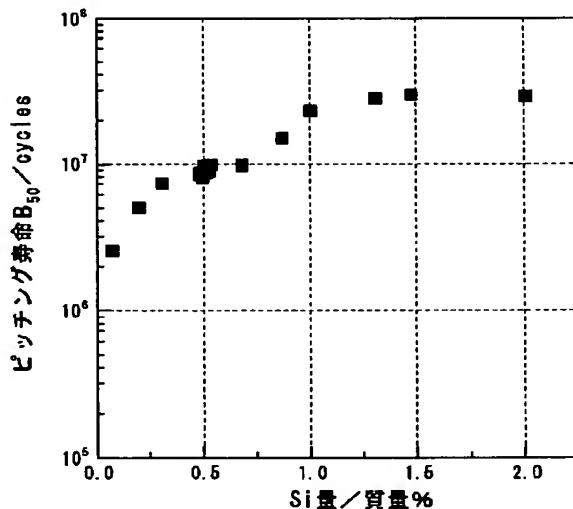
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(54)【発明の名称】 高周波焼入れ用鋼

(57)【要約】

【課題】 耐摩耗性、耐ピッキング性に優れた高周波焼入れ用鋼を提供する。

【解決手段】 合金元素の含有量が、質量%でC: 0.30~0.50%、Si: 0.30~1.50%、Mn: 0.30~1.50%、残部がFeおよび不可避不純物からなり、耐摩耗性および図4に示す耐ピッキング寿命に優れた部品に適用される高周波焼入れ用鋼。



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【特許請求の範囲】

【請求項1】 合金元素の含有量が、質量%で、C:0.30~0.50%、Si:0.30~1.50%、Mn:0.30~1.50%、残部がFeおよび不可避不純物からなり、耐摩耗性、耐ピッキング性に優れた部品に適用される高周波焼入れ用鋼。

【請求項2】 請求項1記載の合金元素に加えて、質量%で、Ni:0.03~3.00%、Cr:0.05~0.30%、Mo:0.01~1.50%の中から選択した1種若しくは2種以上を含有する、耐摩耗性、耐ピッキング性に優れた部品に適用される高周波焼入れ用鋼。

【請求項3】 請求項1または2に記載の合金元素に加えて、質量%で、V:0.05~0.50%、B:0.0005~0.0050%、Ti:0.02~0.05%、Nb:0.005~0.200%の中から選択した1種若しくは2種以上を含有する、耐摩耗性、耐ピッキング性に優れた部品に適用される高周波焼入れ用鋼。

【請求項4】 請求項1~3のいずれか1項に記載の合金元素に加えて、質量%で、Al:0.005~0.050%、N: \leq 0.015%を含有する、耐摩耗性、耐ピッキング性に優れた部品に適用される高周波焼入れ用鋼。

【請求項5】 請求項1~4のいずれか1項に記載の合金元素に加えて、質量%で、S:0.003~0.020%、Pb:0.03~0.20%、Bi:0.03~0.15%、Ca:0.0003~0.0050%の中から選択した1種若しくは2種以上を含有する、耐摩耗性、耐ピッキング性に優れた部品に適用される高周波焼入れ用鋼。

【請求項6】 高周波焼入れ処理後、300°Cで焼戻しをした際のピッカース硬さが500HV以上である請求項1~5のいずれか1項に記載の耐摩耗性、耐ピッキング性に優れた部品に適用される高周波焼入れ用鋼。

【発明の詳細な説明】

【0001】

【発明の属する技術分野】本発明は、耐摩耗性、耐ピッキング性に優れた高硬度部品を得る高周波焼入れ用鋼に関するものである。

【0002】

【従来の技術】これまで自動車のステアリング部品やシャフト等には、JIS SCM440、S45Cに代表される中炭鋼が用いられていた。ところが近年、ステアリングなどにかかる荷重が大きくなっているため、耐摩耗性、耐ピッキング性への要求はますます高まりつつある。特に使用中に高温環境となる場合には、焼戻しにより表面の硬度が減少するため、高温での耐摩耗性、耐ピッキング性が必要となってきているが、その要求は達成されていないのが現状である。

【0003】

【発明が解決しようとする課題】本発明は、上記のような事情を背景としてなされたもので、本発明の目的とするところは、耐摩耗性、耐ピッキング性に優れた高周波焼入れ用鋼を提供することである。

【0004】

【課題を解決するための手段】耐摩耗性、耐ピッキング性向上させるには、焼戻し軟化抵抗を向上させることができるのである。そのため、焼戻し軟化抵抗を向上させるとと言われているSiをJIS SCM440、S4

10 5Cよりも增量添加することにより、JIS鋼と同一の高周波焼入れ条件で処理した場合に、焼戻し軟化抵抗をJIS鋼よりも向上させることが可能で、耐摩耗性、耐ピッキング性が大きく改善されることを見出して本願の発明を得た。

【0005】すなわち、上記の課題を解決するための本発明の手段は、請求項1の発明では、合金元素の含有量が、質量%でC:0.30~0.50%、Si:0.30~1.50%、Mn:0.30~1.50%、残部がFeおよび不可避不純物からなり、耐摩耗性、耐ピッキング性に優れた部品に適用される高周波焼入れ用鋼である。

【0006】請求項2の発明では、請求項1の手段の合金元素に加えて、質量%で、Ni:0.03~3.00%、Cr:0.05~0.30%、Mo:0.01~1.50%の中から選択した1種または2種以上を含有する、耐摩耗性、耐ピッキング性に優れた部品に適用される高周波焼入れ用鋼である。

【0007】請求項3の発明では、請求項1または2の手段の合金元素に加えて、質量%で、V:0.05~0.50%、B:0.0005~0.0050%、Ti:0.02~0.05%、Nb:0.005~0.200%の中から選択した1種または2種以上を含有する、耐摩耗性、耐ピッキング性に優れた部品に適用される高周波焼入れ用鋼である。

【0008】請求項4の発明では、請求項1~3のいずれか1項の手段の合金元素に加えて、質量%で、Al:0.005~0.050%、N: \leq 0.015%を含有する、耐摩耗性、耐ピッキング性に優れた部品に適用される高周波焼入れ用鋼である。

40 【0009】請求項5の発明では、請求項~4のいずれか1項の手段の合金元素に加えて、質量%で、S:0.003~0.020%、Pb:0.03~0.20%、Bi:0.03~0.15%、Ca:0.0003~0.0050%の中から選択した1種または2種以上を含有する、耐摩耗性、耐ピッキング性に優れた部品に適用される高周波焼入れ用鋼である。

【0010】請求項6の発明では、高周波焼入れ処理後、300°Cで焼戻しをした際のピッカース硬さが500HV以上である請求項1~5のいずれか1項の手段に50 おける、耐摩耗性、耐ピッキング性に優れた部品に適用

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される高周波焼入れ用鋼を提供するものである。

【0011】以下に本発明における合金元素の限定理由を説明する。なお、%は質量%で示す。

【0012】C:0.30~0.50%

Cは銅の強度を確保する上で必須の元素であり、その含有量が高周波焼入れ後の硬さを決定する。そこで本発明では、C含有量の下限を0.30%とし、硬さを確保している。しかし、その含有量が多すぎると韌性の劣化や被削性を低下させるなどの弊害をもたらすので、上限を0.50%とする。

【0013】Si:0.30~1.50%

Siは本発明において重要な役割を持つ元素であって、マトリックス中に固溶し、バーライト変態を抑制することにより焼戻し軟化特性を向上させる。その含有量が0.30%未満の場合はその効果が十分には得られず、多すぎると韌性や被削性を低下させるので上限を1.50%とする。

【0014】Mn:0.30~1.50%

Mnは銅の熱間加工性を高め、焼入性を確保する。その含有量が0.30%未満の場合はその効果が十分には得られず、多すぎると被削性や冷鋸性を劣化させて上位を1.50%とする。

【0015】Ni:0.03~3.00%

Niは焼入性を向上させて結晶粒内の強度を増加させる。その含有量が0.03%未満の場合はその効果が十分には得られず、多すぎるとPの粒界偏析を助長するので上限を3.00%とする。

【0016】Cr:0.05~0.30%

Crは焼入性の向上に寄与する。その含有量が0.05%未満の場合はその効果が十分には得られず、多すぎると球状化焼鈍時に炭化物中への濃縮が起こり、高周波焼入れ時に炭化物が残留し、均一な焼入れ硬さが得られないで上限を0.30%とする。

【0017】Mo:0.01~1.50%

Moは焼入性の向上に寄与する。その含有量が0.01%未満の場合はその効果が十分には得られず、多すぎると加工性を劣化させて上位を1.50%とする。

【0018】V:0.05~0.50%

Vは結晶粒微細化に寄与して韌性を向上させる。その含有量が0.05%未満の場合はその効果が十分には得られず、多すぎると機械的性質を劣化させて上位を0.50%とする。

【0019】B:0.0005~0.0050%

Bは粒界強度を強くし、韌性を向上させ、かつ焼入性を

向上させる元素である。その含有量が0.0005%未満の場合はその効果が十分には得られず、多すぎると逆に焼入性を低下させて上位を0.0050%とする。

【0020】Ti:0.02~0.05%

Tiは銅中のfree-Nを固定してBの焼入性への効果を上げ、かつ韌性を向上させる元素である。その含有量が0.02%未満の場合はその効果が十分には得られず、0.05%を超えて銅中のN量が0.015%以下と規定するために、その効果は飽和するので上位を0.05%とする。

【0021】Nb:0.005~0.200%

Nbは結晶粒粗大化を抑制し、韌性を向上させる元素である。その含有量が0.005%未満の場合はその効果が十分には得られず、多すぎるとフェライト地を炭窒化物生成により析出硬化するため、変形抵抗を増加させて上位を0.200%とする。

【0022】Al:0.005~0.050%

Alは脱酸材として必要な元素であり、またNと結合してAINとなり結晶粒の粗大化を抑制する元素である。その含有量が0.005%未満の場合はその効果が十分には得られず、多すぎるとアルミニナ系酸化物が増加して疲労特性、加工性を低下させて上位を0.050%とする。

【0023】N:≤0.015%

Nは0.015%を超えて含有するとTiNが増加し、疲労特性に悪影響を及ぼすので上位を0.015%とする。

【0024】S:0.003~0.020%, Pb:0.

30 0.3~0.20%, Bi:0.03~0.15%, Ca:0.0003%~0.0050%

S、Pb、Bi、Caはいずれも被削性を向上させる元素である。その含有量が少なすぎる場合にはその効果が十分には得られず、多すぎると特性に悪影響を及ぼすのでその範囲を決定する。

【0025】

【発明の実施の形態】本発明の実施の形態の鋼について、表1および表2、図1~4、並びに実施例を通じて示す。表1において、発明鋼1は請求項1の実施の形態、発明鋼2は請求項2の実施の形態、発明鋼3は請求項3の実施の形態、発明鋼4は請求項4の実施の形態、発明鋼5~15は請求項5の実施の形態の鋼を示す。

【0026】

【表1】

(質量%)

	C	Si	Mn	S	Cr	Mo	Ti	B	Al	N
発明鋼	1	0.43	0.49	0.59	0.002	0.03	—	—	—	0.0053
	2	0.47	0.52	0.62	0.001	0.14	0.02	—	—	0.0059
	3	0.44	0.47	0.60	0.002	0.17	0.03	0.033	0.0014	—
	4	0.45	0.49	0.59	0.002	0.15	0.03	0.029	0.0012	0.032 0.0065
	5	0.44	0.31	0.61	0.012	0.14	0.02	0.031	0.0013	0.033 0.0062
	6	0.43	0.51	0.60	0.011	0.14	0.03	0.029	0.0014	0.028 0.0049
	7	0.46	0.69	0.60	0.014	0.16	0.02	0.029	0.0012	0.028 0.0067
	8	0.47	0.88	0.58	0.009	0.17	0.02	0.033	0.0009	0.032 0.0056
	9	0.43	1.01	0.61	0.011	0.14	0.02	0.029	0.0012	0.030 0.0065
	10	0.45	1.32	0.62	0.014	0.15	0.03	0.030	0.0011	0.031 0.0055
	11	0.45	1.49	0.62	0.013	0.13	0.03	0.032	0.0013	0.033 0.0059
	12	0.48	0.53	0.60	0.012	0.28	0.02	0.028	0.0014	0.032 0.0049
	13	0.43	0.51	0.60	0.011	0.13	1.01	0.030	0.0012	0.029 0.0068
	14	0.44	0.49	0.58	0.014	0.16	0.04	0.033	0.0045	0.028 0.0065
	15	0.45	0.50	0.63	0.011	0.19	0.02	0.031	0.0014	0.031 0.0123
比較鋼	1	0.44	0.07	0.59	0.011	0.17	0.02	0.031	0.0011	0.033 0.0055
	2	0.48	0.20	0.60	0.014	0.16	0.02	0.033	0.0018	0.032 0.0068
	3	0.46	2.03	0.58	0.012	0.19	0.03	0.028	0.0013	0.028 0.0063
	4	0.45	0.54	0.62	0.013	0.52	0.02	0.029	0.0013	0.029 0.0060
	5	0.42	0.49	0.63	0.009	0.16	0.02	0.029	0.0099	0.029 0.0057
	6	0.47	0.52	0.61	0.012	0.14	0.02	0.030	0.0012	0.030 0.0211

【0027】

【実施例】表1に示す化学組成の鋼を100kg真空溶解炉にて溶製した後、1150°Cで熱間鍛造して直径32mmの素材を製造した。次いで、前記各素材に対して焼ならしを施した後、図1に示すローラーピッキング試験片1を作成した、このローラーピッキング試験片1は、中央の大径部の直径が26mm、長さが28mmであり、両側の小径部の直径が22mm、長さが51mmの寸法を有するものである。そして、各ローラーピッキング試験片1に対して後記する高周波焼入れ処理を施した後、ローラーピッキング試験を行った。

【0028】高周波焼入れ処理は、出力40kW、周波数200kHz、電圧9.5kV、電流3.2A、コイル移動速度7mm/secの条件で実施し、180°Cで焼戻し処理を行った。

【0029】ここで、ローラーピッキング試験の原理を図2に示す。すなわち、ローラーピッキング試験片1

(小ローラー：材質は表1の鋼)と相手材2(大ローラー：JIS SCM420相当)とを340kgf/m²すなわち、3334N/mm²の面圧下で高速回転させ、ピッキングが発生するまでの寿命である回転数を求めて耐ピッキング性を評価した。なおこの場合、ローラーピッキング試験片1(小ローラー)と相手材2(大ローラー)との周速の差、すなわちすべり率は40%である。そして、各条件についてそれぞれ5本のローラーピッキング試験片1を対象として試験を行った後、ワイルプロットを行い、50%破損確率寿命(B₅₀寿命)を求めて耐ピッキング性を評価した。さらに表面硬化層断

面の表面から100μm位置における硬さを300°Cの高温条件下にて測定した。高周波焼入れ材の結果を表2ならびに図3、4に示す。図3によると、Siの含有量を多くするにつれて300°Cでの硬さが上がっており、0.3%以上の添加で500HVを超えている。なお、Siの含有量を1.5%以上にしてもその効果は飽和している。また、図4によると、Siの含有量を多くするにつれてピッキング寿命B₅₀は向上しており、含有量を1.5%以上にするとその効果は飽和している。

【0030】

【表2】

	300°C高温硬さ(HV)	ローラーピッキング 寿命B ₅₀
発明鋼	1 502	7.8×10^6
	2 514	8.4×10^6
	3 510	8.2×10^6
	4 513	8.0×10^6
	5 501	7.3×10^6
	6 512	8.5×10^6
	7 521	9.6×10^6
	8 531	1.5×10^7
	9 539	2.3×10^7
	10 547	2.8×10^7
	11 554	3.0×10^7
	12 515	8.8×10^6
	13 523	9.8×10^6
	14 519	8.7×10^6
	15 511	7.9×10^6
比較鋼	1 460	2.5×10^6
	2 487	5.0×10^6
	3 556	2.9×10^7
	4 521	9.8×10^6
	5 514	8.3×10^6
	6 510	8.6×10^6

*【0031】

【発明の効果】以上説明したように、本発明鋼はSiを多量に添加することで、焼戻し軟化抵抗を増加させることができ、その結果、高温で硬く、さらに耐ピッキング性に優れた特性を有する部品に適用できる高周波焼入れ用鋼である。

【図面の簡単な説明】

【図1】ローラーピッキング試験片1の概略を示す側面図である。

10 【図2】ローラーピッキング試験の原理を示す斜視図である。

【図3】Si量と300°Cでの硬さとの関係を示すグラフである。

【図4】Si量とピッキング寿命との関係で、高周波焼入れ材のローラーピッキング試験結果を示すグラフである。

【符号の説明】

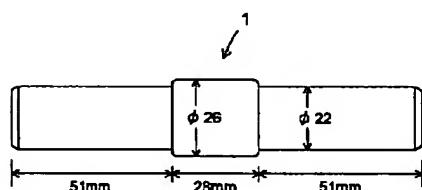
1 ローラーピッキング試験片

2 相手材

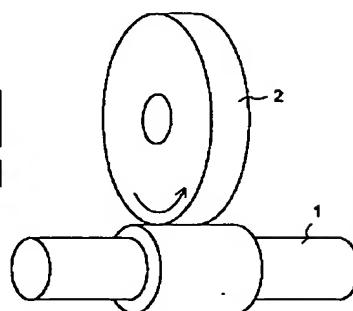
20

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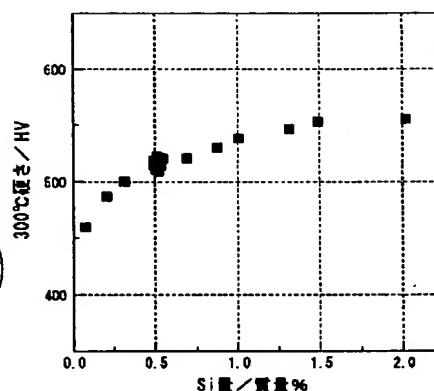
【図1】



【図2】



【図3】



【図4】

